## Fabrication and Evaluation of In Vitro Studies of Biodegradable and Antibacterial Composite Scaffolds Based on Polylactic Acid-Polycaprolactone-Hydroxyapatite Reinforced with Graphene and Zinc Oxide Nanoparticles for Use in Orthopedic Surgery

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#### Abstract:

**Introduction**: Fabrication of fully optimized tissue-engineered materials in order to simulating the natural structure, and enhancing the biological properties of damaged tissue is one of the major challenges in biomedical engineering and regeneration medicine. Although polymeric based membranes have revealed noticeable advancements in bone regeneration, their mechanical stiffens, electrical conductivity and bioactivity need to be tolerated.

**Methods**: Therefore, the present study is designed to generate a multifunctional biomaterial based on polylactic acid (PLA)/ polycaprolactone (PCL)/hydroxyapatite (HA) nanocomposite containing zinc oxide (ZnO) and Graphene (Gr) nanoparticles employing solvent casting combined with die cast techniques for using as absorbable joint implants in bone tissue regeneration. The physical, chemical, mechanical and biological properties of the produced nanocomposite biomaterials were analyzed in vitro. A detailed experimental evaluation between the nanocomposite coatings was carried out to shed light on the effect of ZnO and Gr nanoparticles on the properties. **Results**: It was found that the nanocomposite contained 1% ZnO and 1% graphene with a Young's modulus of  $1540.5 \pm 169.426$ MPa and the pure sample had a Young's modulus of  $1194.81\pm215.342$ MPa. The rate of elongation at break of the nanostructure contained 1% graphene was  $5.1\pm0.816\%$ . This value was  $3.8\pm0.944\%$  for the pure sample. The improvement in elongation at break is due to the presence of polycaprolactone in the polymer matrix. The optimal sample with 1% zinc oxide and 1% graphene had antibacterial properties more than other samples. Also, the survival rate of fibroblasts cell in the vicinity of the optimal matrix was significantly different from other samples.

**Conclusion**: The obtained results revealed that the incorporation of the nanoparticles improved physico-chemical features and mechanical strength with enhanced biological properties and its anti-bacterial performance makes this material a promising candidate for further bone regeneration studies.

*Keywords:* Polycaprolactone, polylactic acid, hydroxyapatite, graphene, zinc oxide.

#### **1. INTRODUCTION**

Up to now, bone damages and failures as well as related defects have been recognized as a serious health problem by world health organization (WHO) as a result of dramatical rise in the bone diseases such as bone tumor, ageing trauma population, damages from sport activities and traffic accident (1). From realistic point of view, new advancements in fully engineered tissues have offered an alternative method to deal with bone related disease through altering damaged multifunctional sites with substitution in comparison with traditional approaches including allografts and autografts, introducing highly potential solutions to overcome the shortages of donated tissues (2-4). In addition, traditional therapies suffer from some disadvantages including disease transmission, need for further surgery, graft rejection, and pain (5). tissue (TE) nanocomposite coatings engineering support attachment, migration, and proliferation



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of host cells (6). Current studies shown that many approaches and biomaterials have been tested such as natural or synthetic biomaterials, stem cells from different sources, and grafts in order to restore/ regenerate massive defects, designing a suitable three-dimensional (3D) substrate to eliminate disadvantages of the grafts (7-9). The TE nanocomposite coating is a supporting place for attachment, migration, proliferation, and differentiation of cells, and its main function is reproducing the ECM. Thus, its role to control cell behaviors and fates can result in naturally tissue regeneration (10).

Recently, various types of bio-materials have been employed in fabrication of tissue-engineered composite coating to enhance bone regeneration. For instance, He et al. fabricated collagen/ poly lactide (PLA)/ nano-hydroxyapatite composite coating to induce mesenchymal stem cells (MSCs) differentiation (11). Moreover, Oryan et al. produced β-tricalcium phosphate/ chitosan membrane to enhance bone reconstruction (12). The mentioned nanocomposite coating are designed with different types of biomaterials such as synthetic, natural, or a combination of both. It is noted that the nanocomposite coating should be with suitable degradation designed rate. mechanical stiffness, biocompatibility, and the nanocomposite coating used for TE should degrade at a rate that tuned with the rate of tissue growth (13). Polycaprolactone (PCL) and PLA as synthetic polymers have been applied to regenerate bone structure due to their mechanical properties, biodegradation rate. and biocompatibility. PLA is fragile with low elongation at break, thus, its combination with polymers such as PCL which is a stiffness polymer with higher elongation at break properties can enhance its mechanical features. On the other hand, cell/scaffold interaction of PCL is improved with the addition of PLA due to surface energy (14-16). Therefore, better fabricated composite coating can show better properties in comparison with the single polymer, however, there is still room for improvement because of its lack of functional groups and hydrophobicity in the PCL/PLA composite which need to be modified for biomedical applications. Additionally, the nanocomposite coating for bone regeneration needs having proper osseointegration and osteoinductivity potential (17). In addition, other features such as

mechanical properties, hydrophilicity and biocompatibility would be improved with the combination of bioactive agents such as hydroxyapatite (HA). The addition of HA also enhances bone cells attachment, proliferation, and differentiation. Furthermore, mineralization and bone formation is increased due to its calcium components (18). Currently, various types of nanoparticle have been used to fabricate nanocomposite (19, 20). According to diverse types of nanofillers, using inorganic nanoparticles can be a wise selection because of their antimicrobial and mechanical strength properties (21, 22). Furthermore, one of the most crucial factor in the fabrication of nanocomposite with using nanoparticles is dispersion of inorganic types in the polymer matrices because it can improve composite properties drastically, which is possible with using ZnO due to its particleparticle interactions and surface energy (23). Zinc oxide (ZnO) which is approved by the Food and Drug Administration (FDA) in addition to its functional properties semiconductor is nanoparticle which can be used because of its low toxicity, chemical stability, anti-bacterial capacity and influential oxidation strength, and it is lowcost and it is known as a multifunctional inorganic nanoparticle (24-27). Moreover, graphene (Gr), a single atomic membrane, is recognized as the most versatile biomaterial due to its electrical, mechanical, optical, and thermal properties and high surface area which make this a promising material is biomedical applications (28).

Regarding the previous studies, bone tissue-like structure was fabricated by different techniques. such as electrodeposition, die cast, dip coating, spin coating, electrospinning, layer by layer assembly, freeze drying, solvent casting, covalent linkage, co-precipitation, and ionic interaction techniques (29-31). Among the above approaches, solvent casting and die casting simplest, are two methods, the specific approaches which are solution-based techniques to fabricate membrane or film for simulation of the tissue microenvironment structure (32).

In the present study, casting after solution mixing and injection molding were used in order to producing composite with different concentrations of Gr and ZnO nanoparticles as filler materials incorporating in PLA/ PCL/HA composite. The physical, chemical, and mechanical properties the fabricated of



nanocomposites were evaluated via FESEM, EDS, FTIR, XRD, contact angle, mechanical evaluation, electrical conductivity, and In vitro degradation. Then, biological properties of fabricated nanocomposite coating were analyzed, and antibacterial assessment was carried out.

## 2. MATERIALS AND METHODS

## 2.1. nanocomposite coating fabrication

The PLA/PCL, PLA/PCL/HA, PLA/PCL/HA/ZnO, and PLA/PCL/HA/ZnO/Gr membranes were fabricated via solvent casting method combined with a die cast technique (Table 1). In brief, fabrication of PPH0Z0G0 which was chosen as a control group was carried out as follows: the combination of 5 g PLA and 1.2 g PCL (80/20 w/v, 80000 g/mol, both purchased from Sigma-Aldrich) was dissolved in Chloroform (Cl, 90 mL/g, Sigma-Aldrich). After 24h under magnetic stirrer condition, the obtained solution was sonicated for 30 min at room temperature. As it is shown in table 1, fully dissolved PLA/PCL solution was further mixed and sonicated with HA/Cl (3 wt.%, Parsian Merck, Iran) solution to obtain a clear and homogenous solution. Addition of ZnO and Gr (both purchased from Parsian Merck, Iran) nanoparticles were carried out similarly. In brief, PLA/PCL, and HA were combined with Cl (3 wt.%) separately. ZnO and Gr nanoparticles also mixed with Cl (3 wt.%) separately. Finally, all thoroughly mixed solutions were combined followed by using sonication for 30 min.

The membrane was fabricated through using solvent casting technique, then die casting membranes was produced, the prepared mixtures were purred onto glass petri-dishes after evaporation of solvent at room temperature (37°C), and vacuum oven was used for completely removed excess solvent for 48 h, further stored for further studies.

## 2.2. Morphological studies

To detect and evaluate the incorporation of nanoparticles into the films plus the nanoparticles surface area, pore size, pore volume, and cross section micrographs were visualized with field emission scanning electron microscope (FESEM, TESCAN-Vega 3, Czech Republic) equipped with energy dispersive spectroscopy (EDS) at 20 kV. Before morphological assessment, the

prepared films mounted on the brass stub and sputtered coated under a flow of argon to enhance the conductivity of the samples.

## 2.3. Mechanical strength

Regarding the standard ASTM-D638, mechanical properties of fabricated films were determined via a Texture Analyzer TA. XT plus (TM-SM, Instron, UK) by a 50 N load cell under a crosshead speed of 50 mm/min. the tensile strength and elongation at break were evaluated in 5 specimens from each of the produced films. Fabricated nanocomposite coatings also were evaluated in the wet state to observe the mechanical response into in vivo condition. To reach this purpose; the mechanical test was carried out after soaking the films in PBS for 48h. each sample were assessed after five times repetition.

## 2.4. Chemical structure investigation

The functional groups of the prepared membranes were investigated via Fourier transform infrared spectrometer (FTIR, Nicolet, NEXUS 670, USA) In the range of  $400-4000 \text{ cm}^{-1}$ .

## 2.5. Hydrophilicity Studies

To observe the hydrophilicity of different samples, contact angle meter (XCA-50) was used. In brief, samples were analyzed and reported after 15s of water dropping through using a water droplet with a volume of 4 µL. The reported results were based on the average of five times repetition.

## **2.6. Electrical conductivity**

To evaluate the electrical conductivity of the nanocomposite coatings, four steel blades were placed on the membranes. The associated conductivity ( $\delta$ ) was accomplished based on Pouillet's law (28) as follows (Eq. 1): resistance (R) and Electrical conductivity (EC) were assessed five times for different samples individually.

$$\delta = L/(R \times A) \tag{1}$$

where A, and L were considered as the surface area, and length of the fabricated samples, respectively.

## 2.7. Characterization of the nanoparticles and films

To assess nanoparticles incorporated PPH0Z0G0, PPH1Z0G0, and PPH1Z0.5G0.5 membranes, structural analysis was accomplished via X-ray diffraction (XRD, STOE, IP-PSD).



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#### 2.8. In vitro degradation

To measure the biodegradation behavior of the membranes, the fabricated nanocomposite coatings (n=5) were similarly cut ( $5 \times 5 \times 3 \text{ mm}^3$ ), weighed  $(W_0)$ , and soaked in PBS (Ph = 7.4), and after that the incubation process was carried out at 37 °C with continuous shaking at 50 rpm for 14 days. This should be mentioned that the PBS was changed at each interval time point. To assess the weight loss, at different time point (3, 7, and 14 samples were transferred days), the for elimination of excess water, next rinsed with distil water (DW), then dried at vacuum oven at 37°C and weighed (W<sub>d</sub>). The calculation of weight loss was carried out regarding the following equation (Eq. 2) (33):

 $\Delta W (\%) = \frac{W_0 - W_d}{W_0} \times 100\%$ (2)

#### 2.9. Calcium and zinc release

Ca and Zn ions release from the nanocomposites was evaluated by using coupled plasma-optical emission spectrum (ICP-OES, Pekin Elmer-Optima 7300 DV). The nanocomposites were cut into similar dimensions, then soaked into SBF (10 mm, n=5). After that, the ion release profile was observed in 3, 7, and 14 days.

#### 2.10. Bioactivity evaluation

To observe the bioactivity of the nanocomposite coatings, fabricated membrane with equal dimension  $(5 \times 5 \text{ mm}^2)$  was placed into completely sterilized petri dishes, and then soaked in SBF (Ph = 7.4) at room temperature for four weeks. After that, at each preselected time, they were dried at 37 °C, and then analyzed with SEM and XRD to observe the formation of HA crystal, and its depositions. In addition, the amount of Ca, Zn, and P ions was calculated via ICP-OES in the SBF.

#### **2.11. Biological performance**

Mouse fibroblast (L929, Pasteur, Iran) was used as a cell line. In order to cell culture, briefly, the collected cells were seeded in Dulbecco's Modified Eagle Medium/Nutrient Mixture F-12 (DMEM-F12, Gibco, USA) medium, 1% antimycotic and antibiotic formulation, 1% Lglutamine (all from Gibco, USA), with 10% fetal **BIOCHROM** bovine serum (FBS; AG, Germany), containing amphotericin B, streptomycin sulfate, and penicillin G sodium (all from Invitrogen). After covering plates with cells, the cultured cells were rinsed two times with 4 mL PBS. After that, detachment of cells was carried out by EDTA/trypsin, then the cells were centrifuged (1200 rpm, 4 min). The obtained cells were cultured and incubated again. The culture medium (37 °C in a humidified condition with 5%  $CO_2$ ) was changed every two days.

The cytotoxicity assessment (MTT assay, Sigma-Aldrich) was carried out to observe lived-dead cells. First, the membranes were cut into disk shapes equally, then placed into 24-well plate. Then, the samples were placed into incubator (37°C for 24 h). After that, they (L929 with a density of  $10 \times 10^3$ ) were cultured, and incubated onto each sample. It should be noted that the medium was changed at each 3-4 days. To evaluate MTT assessment, the culture medium was removed, and the samples were washed by PBS three times. Next, 50 µL MTT and 500 µL medium was added to each plate. after incubation (37°C for 3-4 h), the medium was extracted, and 400 µL DMSO (Sigma-Aldrich) was replaced to the wells. Then, the obtained solution was pipetted and placed in the dark condition (depended on the detachment rate of cells). as the final step, Elisa reader (570 nm wavelength, Bio-Rad, C55, Germany) was employed to assess with transferring 100 µL of each sample to 96-well plate (n=5) based on revived formazan in 24 and 72 h. The cell viability (%) was evaluated by Eq. (3):

Cell viability (%) = 
$$\frac{OD_{sample}}{OD_{control}} \times 100\%$$
 (3)

## 2.12.In vitro anti-microbial assessment

The antimicrobial efficiency of the nanocomposite membranes was assessed against E. coli and S. aureus as Gram-negative and Grampositive, respectively, with association of the zone of inhibition technique (34). Briefly, in vitro antimicrobial evaluation, a multi-step protocol was carried out one after another as follows: (1) dipping the membranes; (2) incubation of the inoculums; (3) shaking the tubes at room temperature; (4) extracting the aliquots; (5) diluting the samples serially by PBS and then spreading onto the plates; (6) incubation of the plates at room temperature for 24h. Finally, the colony-forming units were calculated. PLA/PCL was used as a control sample.

#### 2.13. Statistical analysis

All reported experimental data were arranged on mean and standard deviation ( $\pm$ SD). In addition, SPSS software and a one-way ANOVA test were



used to assess cellular data. It should be mentioned that,  $p \le 0.05$  was recognized statistically substantial.

#### **3. RESULTS AND DISCUSSIONS**

# **3.1.** PLA / PCL / HA / ZnO / Gr nanocomposite coating fabrication

Over this section. different steps of described. nanocomposite fabrication are Producing PLA/PCL/HA/ZnO/Gr with optimized properties was the main purpose, however this needs step by step evaluation. Thus, our first goal was to explore a proper weight ratio between PLA/PCL/HA and ZnO, and Gr to produce a nanocomposite coating with sufficient mechanical, chemical, morphological, physical, and biological properties because the mentioned nanoparticles and polymers are introduced biocompatible for tissue engineering applications. First step was optimization of ZnO with PLA/PCL/HA through three weight ratiosas described in Table 1 in addition HA was evaluated for the combination of PLA/PCL to find which one would generate the nanocomposite with promising properties. The FESEM micrographs and cross-sections of fabricated nanocomposite coatings are shown in Fig. 1 and Fig. 2, where the effect of HA, ZnO, and Gr addition to the PLA/PCL membrane is obvious. Furthermore, the textural properties pore size (µm) and mechanical properties elongation (%), ultimate tensile strength (UTS, MPa), young's modulus (MPa), and wettability of fabricated membranes are shown through Table 2 and 3, respectively. According to the Fig. 1 and Table2, the PLA/PCL nanocomposite revealed homogenous and similar

interconnected pore size ~ 1.52 µm. With combination of HA, the pore size was increased up to  $\sim 1.82 \,\mu m$ , increasing surface flexibility without risk of early degradation, which promote membranes adoptability features for further studies (35). As it is visible, the addition of ZnO affects the morphological properties, raising in the nanoparticles content up for PPH1Z0.1G0, PPH1Z0.5G0, and PPH1Z1G0 promoted pore size to  $\sim 2.51$ , 2.84, and 3.66  $\mu$ m, which may be the consequences of reducing interaction between polymers and ZnO, which is normal in the combination of nanoparticles into membranes (36). Then, Gr incorporation further showed better distribution and encouraging up for PPH1Z0.1G0.1, PPH1Z0.5G0.5, and PPH1Z1G1 to ~ 3.01, 3.27, and 4.01  $\mu$ m, and its morphological properties were proper in shape. Our major focus was based on using the maximum percentages of nanoparticles because of positive effect on mechanical, chemical, and biological properties without any cytotoxicity. Therefore, our further investigations were based on blending 1 wt.% ZnO and 1 wt.% Gr with the prepared membrane. Fig. 2 illustrated the pores of prepared nanocomposite coatings were interconnected, and their shapes were irregular in structure with different properties. However, the addition of ZnO and Gr affects positively in the structure with good textural properties which could revealed proper mechanical, chemical and biological features.

Appropriate mechanical performance such as compressive strength, fatigue strength, tensile strength, shear modulus, toughness and young's modulus are the most noticeable features in bone tissue regeneration (37).

| Sample       | Composition       | PLA/PCL<br>(w/v) | Hydroxyapatite<br>(w/v) | Zinc<br>oxide<br>(w/v) | Graphene<br>(w/v) |
|--------------|-------------------|------------------|-------------------------|------------------------|-------------------|
| PPH0Z0G0     | PLA/PCL           | 80/20            | _                       | _                      | _                 |
| PPH1Z0G0     | PLA/PCL/HA        | 80/20            | 1 %                     | _                      | _                 |
| PPH1Z0.1G0   | PLA/PCL/HA/ZnO    | 80/20            | 1 %                     | 0.1 %                  | _                 |
| PPH1Z0.5G0   | PLA/PCL/HA/ZnO    | 80/20            | 1 %                     | 0.5 %                  | -                 |
| PPH1Z1G0     | PLA/PCL/HA/ZnO    | 80/20            | 1 %                     | 1 %                    | -                 |
| PPH1Z0.1G0.1 | PLA/PCL/HA/ZnO/Gr | 80/20            | 1 %                     | 0.1 %                  | 0.1 %             |
| PPH1Z0.5G0.5 | PLA/PCL/HA/ZnO/Gr | 80/20            | 1 %                     | 0.5 %                  | 0.5%              |
| PPH1Z1G1     | PLA/PCL/HA/ZnO/Gr | 80/20            | 1 %                     | 1 %                    | 1 %               |
| PPH1Z1G2     | PLA/PCL/HA/ZnO/Gr | 80/20            | 1 %                     | 1 %                    | 2 %               |

Table 1. Composition of prepared membranes.







**Fig. 1.** FESEM micrographs of PPH0Z0G0 (A), PPH1Z0G0 (B), PPH1Z0.1G0 (C), PPH1Z0.5G0 (D), PPH1Z1G0 (E), PPH1Z0.1G0.1 (F), PPH1Z0.5G0.5 (G), and PPH1Z1G1 (H) prepared membranes. Scale bar was 20 μm.



**Fig. 2.** Cross section of PPH0Z0G0 (A), PPH1Z0G0 (B), PPH1Z0.1G0 (C), PPH1Z0.5G0 (D), PPH1Z1G0 (E), PPH1Z0.1G0.1 (F), PPH1Z0.5G0.5 (G), and PPH1Z1G1 (H) prepared membranes. Scale bar was 20 μm.

| Table 2. | The textural properties of PLA- | -PCL, PLA-PCL-HA, PLA-PCL | L-HA-ZnO, and PLA-PCL-HA-ZnO-Gr |
|----------|---------------------------------|---------------------------|---------------------------------|
|          |                                 |                           |                                 |

| membranes    |                  |  |  |  |
|--------------|------------------|--|--|--|
| Sample       | Pore size (µm)   |  |  |  |
| PPH0Z0G0     | $1.52 \pm 0.241$ |  |  |  |
| PPH1Z0G0     | $1.82 \pm 0.146$ |  |  |  |
| PPH1Z0.1G0   | $2.51 \pm 0.501$ |  |  |  |
| PPH1Z0.5G0   | $2.84 \pm 0.257$ |  |  |  |
| PPH1Z1G0     | $3.66 \pm 0.71$  |  |  |  |
| PPH1Z0.1G0.1 | $3.01 \pm 0.917$ |  |  |  |
| PPH1Z0.5G0.5 | $3.27 \pm 1.122$ |  |  |  |
| PPH1Z1G1     | $4.01 \pm 0.564$ |  |  |  |

Sufficient mechanical behavior to withstand stress after implantation (maintaining its features during the surgical process), and keeping its microstructure during of new bone formation.

Therefore, having enough mechanical properties in wet state plus dry state are of great importance (36). The mechanical properties of PLA/PCL incorporated nanoparticles are summarized in



Fig. 2 and Table 3 in both dry condition and after soaking membranes into SBF, representing that incorporation of nanoparticles enhanced tensile strength significantly.

The enhancement of young's modulus is generally associated with lower interfacial energy between the nanoparticles and polymer matrix (38, 39). Although nanoparticles incorporation resulted in improving mechanical properties in dry and wet state, as it was revealed its Young's modulus was decreased with addition of 2% Gr, which could be because of agglomeration effect (40). Thus, PPH1Z1G1 nanocomposite presented the best mechanical performance among different percentages of nanoparticles in terms of UTS=  $20.91 \pm 1.001$  MPa and young's modulus=  $1742.5 \pm 316.541$  MPa in dry state, and UTS=  $13.17 \pm 0.458$  and young's modulus=  $1097.5 \pm 71.412$  in wet state.

As synthetic materials are known as a material with hydrophobic components, this feature is valued disadvantage as far as natural bone tissue reconstruction is needed (17). To solve this, there have been some approaches such as surface treatment, adding bioceramics, and combination with natural polymers can be used.



Fig. 3. The stress-strain curves of prepared membranes in dry (A) and wet (B) conditions.

| and I LAT CE-TIA-Eno-Gi inclusiones. |                         |               |             |              |   |               |                  |
|--------------------------------------|-------------------------|---------------|-------------|--------------|---|---------------|------------------|
| Sample                               | Young's Moudulus<br>(%) |               | UTS(Mpa)    |              | Elongation (%)                                  |               | Water<br>contact |
| _                                    | Dry                     | Wet           | Dry         | Wet          | Dry   | Wet           | angle (°)        |
| PPH0Z0G0                             | $450 \pm$               | $350.24 \pm$  | $9\pm0.538$ | $8.12 \pm$   | 5.1 ±   | $4.9 \pm$     | $89 \pm$         |
| FFHUZUGU                             | 87.176                  | 52.102        | 9±0.338     | 0.301        | 0.417   | 0.311         | 0.103            |
| PPH1Z0G0                             | $648.5 \pm$             | $542.83 \pm$  | $12.97 \pm$ | $10.58 \pm$  | 4.7 ±   | 4.5 ±         | $86 \pm$         |
| FFHIZUGU                             | 92.613                  | 23.412        | 0.451       | 0.249        | 0.339   | 0.389         | 0.271            |
| PPH1Z0.1G0                           | $720.5 \pm$             | $590.12 \pm$  | $13.7 \pm$  | $11.23 \pm$  | $\begin{array}{c} 5.2 \pm \\ 0.408 \end{array}$ | $5 \pm 0.409$ | $86 \pm$         |
| FFHIZ0.IG0                           | 91.312                  | 31.175        | 0.508       | 0.327        |   |               | 0.569            |
| PPH1Z0.5G0                           | $953.33 \pm$            | $680.75 \pm$  | $14.41 \pm$ | $10.28 \pm$  | $5\pm0.412$                                     | $4.9 \pm$     | $83 \pm$         |
| 111120.300                           | 93.981                  | 36.567        | 0.672       | 0.238        |   | 0.389         | 1.329            |
| PPH1Z1G0                             | $1050.66 \pm$           | $782.661 \pm$ | $15.76 \pm$ | $11.74 \pm$  | 5.3 ±   | 5.2 ±         | $75 \pm$         |
| 11112100                             | 104.052                 | 43.734        | 0.717       | 0.389        | 0.441   | 0.341         | 0.153            |
| PPH1Z0.1G0.1                         | $1017.64 \pm$           | $678.45 \pm$  | $17.3 \pm$  | $11.533 \pm$ | $6.2 \pm$                                       | $6.1 \pm$     | $84 \pm$         |
| 1111120.100.1                        | 112.719                 | 38.378        | 0.919       | 0.372        | 0.628   | 0.506         | 2.181            |
| PPH1Z0.5G0.5                         | $1260.66 \pm$           | $854.54 \pm$  | $18.91 \pm$ | $12.81 \pm$  | $6.5 \pm$                                       | $6.3 \pm$     | $90 \pm$         |
| 111120.360.3                         | 213.306                 | 62.701        | 0.837       | 0.439        | 0.672   | 0.595         | 2.454            |
| PPH1Z1G1                             | $1742.5 \pm$            | $1097.5 \pm$  | $20.91 \pm$ | $13.17 \pm$  | $5.3 \pm$                                       | $5.3 \pm$     | $78 \pm$         |
|                                      | 316.541                 | 71.412        | 1.001       | 0.458        | 0.462   | 0.468         | 1.232            |
| PPH1Z1G2                             | $1350 \pm$              | $981.21 \pm$  | $19.5 \pm$  | $12.1 \pm$   | 5.2 ±   | $5\pm0.371$   | $114 \pm$        |
| 11112102                             | 240.112                 | 69.543        | 0.994       | 0.389        | 0.451   |               | 3.651            |

Table 3. Mechanical properties and wettability of PLA-PCL, PLA-PCL-HA, PLA-PCL-HA-ZnO,and PLA-PCL-HA-ZnO-Gr membranes.



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To measure the effect of nanoparticles addition on the hydrophilicity of PLA/PCL nanocomposites, the wettability test was considered. The water contact angle (WCA) data is shown in Table. 3. The WCA of PLA/PCL was about 89°. The combination of HA with polymeric matrix enhanced hydrophilicity to about 86°. The further measurement showed that the WCA of PPH1Z0G0, PPH1Z0.1G0 and PPH1Z0.5G0 were about 86, 83 and 75°, respectively. Although addition of ZnO improved wettability of nanocomposites because it resulted in the larger micro-pores, its hydrophilicity improvement was not significant (41). After that, the hydrophilic properties of nanocomposite coatings also improved with addition of Gr, proving sufficient performance with increasing amount of Gr (42). Results demonstrated that well-dispersed ceramic nanoparticles in scaffold improved hydrophilicity and mechanical properties compared with agglomerated ceramic nanoparticles. In this research agglomeration has happen in the structure of PPH1Z1G2 scaffold (2% Gr), that caused to decrease mechanical properties and increase hydrophobicity (6, 43).

Regarding the mentioned properties of ZnO/Grincorporated PLA/PCL/HA nanocomposite coating, it was exposed that PPH1Z1G1 showed outperforming properties than the others in chemical, morphological, and mechanical stability. Therefore, PLA/PCL/HA with incorporation of 1 wt.% ZnO and 1 wt.% Gr was considered as an optimized sample and the cell performance-bioactivity, compatibility biodegradation, and cell viability-were evaluated experimentally and compared with of PLA/PCL PLA/PCL/HA those and nanocomposite samples.

**3.2. The PLA / PCL / HA (1 wt.%) /ZnO (1 wt.%) / Gr (1 wt.%) nanocomposite coating** The chemical groups of different membranes - PLA/PCL, PLA/PCL/HA, PLA/PCL/HA/ ZnO, PLA/PCL/HA/ ZnO/Gr - were evaluated through FTIR and the collected results are shown in Fig. 4, where the FTIR spectrum of PCL correspond to the CH<sub>2</sub> asymmetric and symmetric, and carbonyl stretch, C-O stretch were at 2860, 2940, 1720, and 1271 cm<sup>-1</sup>; while, at 1720, 2883, and 1032 cm<sup>-1</sup> represent carbonyl stretch, CH3 stretch, and C-O stretch of PLA, respectively. Moreover, Fig. 4(b) represented the spectrum of HA where PO4<sup>-3</sup> stretch band, and OH stretch was around 974 and  $3500 \text{ cm}^{-1}$ , respectively (10, 44). Fig. 4(c) bands at 1370 cm<sup>-1</sup> showed symmetric and 1600 cm<sup>-1</sup> illustrated asymmetric stretches. However, additional peak was not revealed after mixing ZnO into membrane as it was reported earlier by Shankar et al. and Huang et al. for PLA/ZnO film (41, 45, 46). Functional groups of Gr also appeared at 1530 and 1550 cm<sup>-1</sup> proves represented the vibrational band G and asymmetric tensile vibration of the C-C bond, respectively (47-49).

The XRD evaluations of PLA / PCL / HA, PLA / PCL, and PLA / PCL / HA / ZnO / Gr membranes are revealed in Fig. 5. Illustrating the XRD peaks of ZnO and Gr nanoparticles hither was due to proving completely distinguishing the nanoparticles in the nanocomposite's pattern. Two distinguishable peaks of PLA and PCL were observed in the range at 16.5° and 18.8° (50). The mentioned peaks also can be seen in XRD observation PLA/PCL/HA of ternary nanocomposite with some changes. However, the distinguishable peaks at 26°, 31.63°, 32.42°, and 33.25° through hexagonal shapes can prove HA composition (51). The crystalline phase of PLA/PCL/HA/ZnO/Gr nanocomposite was assessed via X-ray diffraction test. The PLA/PCL/HA are shown with some changes. The distinguishable peaks of ZnO were observed at 31.9°, 33.4°, 47.1°, 56.1°, 63.1° (52), In addition, peaks at 25.8° (002) and at 42.7° (101) in Fig. 5, were assigned to f hexagonal wurtzite geometry in ZnO (JCPDS 36-1451) (53) . and the diffraction peaks of Gr at 10.69°, while no other crystalline peaks were observed related to Gr (54).

Fabricating nanocomposite with the best chemical, mechanical, physical, and biological properties is the foremost reason of incorporation of nanoparticles into polymeric components (55). From materialistic point of view, nanoparticles such as ZnO and Gr are well-known to enhance the cell/scaffold interaction like cell adhesion, proliferation, and growth because of offering some functional ions in order to simulating native ECM.





Fig. 4. The FTIR spectra of PLA-PCL (A), PLA-PCL-HA (B), PLA-PCL-HA-ZnO (C), and PLA-PCL-HA-ZnO-Gr (D).









**Fig. 6.** the FESEM micrographs relating to bioactivity potential of PLA-PCL (A), PLA-PCL-HA (B), PLA-PCL-HA-ZnO (C), and PLA-PCL-HA-ZnO-Gr (D), and EDS analyses of PLA-PCL-HA-ZnO-Gr (E, F, G, and H).

Furthermore, guaranteeing osteointegration with the host targeted sites is considered as an ideal bone membrane because this tissue-engineered nanocomposite coating can bring active environment to form calcium phosphate layer on

the surface. The generated layer will impact positively to make a functional bond between the nanocomposite coating and host bone. Thus, bone tissue regeneration will be happening (56). One of the most reliable ways to simulate the formation



of calcium phosphate layer of fabricated nanocomposite coatings is to soak them into the SBF over some preselected times (57). In the present study, this method was carried out to evaluate the apatite nucleation and growth potential of PLA/PCL, PLA/PCL/HA, PLA/PCL/ZnO/Gr PLA/PCL/ZnO, and membranes up to 60 days experimentally. After 60-days submerging, nanocomposite coatings were evaluated through SEM micrographs. Furthermore, the elemental assessment was considered to prove calcium phosphate newly formation. The obtained results are involved the SEM micrographs of PLA/PCL, PLA/PCL/HA, PLA/PCL/HA/ZnO, and PLA/PCL/HA/ZnO/Gr, and EDS analyses of PLA/PCL/HA/ZnO/Gr nanocomposite are illustrated in Fig. 6. As it is obvious, with increasing time up to 60 days, calcium phosphate was formed on different nanocomposite coatings, especially on the PPH1Z1G1 surface. Interestingly, the calcium phosphate layer of PLA/PCL/HA/ZnO/Gr was totally formed on the surface. In addition, the EDS test proved that nanocomposite coatings were covered mostly with phosphorus and calcium, which showed calcium phosphate formation on the PLA/PCL/HA/ZnO/Gr nanocomposite coating. The collected results of bioactivity assessment revealed that the nanoparticles loaded in PLA/PCL membrane inducing the bioactivity kinetic resulted in substantially.

According to current studies, the cell/scaffold interaction can be improved with increasing the electrical conductivity of fabricated nanocomposite coatings (58). Moreover, addition of electroconductive nanoparticles prevents the agglomeration of nanoparticles, improving chemical and physical properties of membrane. Thus, electrical conductivity of fabricated membranes was investigated as described above, and Table 4 presented addition and higher amount of Gr is a main reason to enhance the electrical properties. The release profile showed that, after the nanocomposite coating was placed into SBF solution, Ca ions were released. The results at each interval times are shown in Fig. 7. First and foremost, the nanoparticles weight ratio PLA/PCL/HA loaded into nanocomposite coatings were estimated. To weight of fabricated calculate this, the membranes was fixed about 2 mg. Thus, the weight of Ca and Zn were approximately 450, and 550 µg for PPH1Z1G0 and PPH1Z1G1, respectively. After that, the membranes were weighed after removing from the solution through ICP. The release profile was assessed for both Ca and Zn ions separately. For Ca ion, after one week the rate of release was about 91, 93.5, and 74 mg/l, and for Zn ion was about 0.01, 8.5, and 8.6 mg/l for PPH1Z0G0, PPH1Z1G0 and PPH1Z1G1, respectively. Moreover, the Ca ion release was raised in duration of 14 days for both PPH1Z1G0 and PPH1Z1G1, however, it was more noticeable for PPH1Z1G1 (about 176 mg/l). Furthermore, Zn release for PPH1Z1G0 was more substantial (about 147 mg/l) in comparison with PPH1Z1G1, which was increased steady.

One of the most noticeable properties of tissue engineered nanocomposite coating for bone tissue regeneration is having adjustable in vitro degradation, and this must to be tuned with bone newly formation, which has a noticeable impact on nanocomposite coating applicability (59). The in vitro degradation of fabricated nanocomposites including PLA / PCL, PLA / PCL / HA, PLA/PCL/HA/ZnO, PLA/PCL/HA/ZnO/Gr was evaluated into PBS aqueous solution up to 20 weeks. It is proved that PLA/PCL showed biodegradation rate slowly which can be found in Fig. 8. After HA blending into PLA/PCL nanocomposite coating, the rate of degradation was increased due to improving hydrophilicy of membrane. It should be mentioned that degradation rate of the nanocomposite coating like pure PLA/PCL as a control sample can be used to observe the positive effect of combination of HA into nanocomposite coating because the nanocomposite coating had degraded approximately 6.48 % of its weight up to 20 weeks.

 Table 4.
 Electrical conductivity of PLA-PCL, PLA-PCL-HA, PLA-PCL-HA-ZnO, and PLA-PCL-HA-ZnO-Gr

| Sample   | Electrical conductivity (S/M) | probability |
|----------|-------------------------------|-------------|
| PPH0Z0G0 | $0.00046 \pm 0.00097$         | 0.016       |
| PPH1Z0G0 | $0.00055 \pm 0.0007$          | 0.017       |
| PPH1Z1G0 | 0.007122 ±0.0013              | 0.007       |
| PPH1Z1G1 | $0.010416 \pm 0.012$          | < 0.005     |





Fig. 7. The Zn (A) and Ca (B) release profile of PLA-PCL-HA, PLA-PCL-HA-ZnO, and PLA-PCL-HA-ZnO-Gr.

After careful evaluation of PPH1Z1G0 and PPH1Z1G1 nanocomposite coatings, the *in vitro* degradation of the nanocomposite coatings was increased to about 8.92 and 9.31 % after the end of period, respectively. It can be concluded that after soaking them into PBS, the nanoparticles were undergone this process in addition of PLA/PCL, and the nanocomposite coatings revealed a continuous and higher biodegradation rate than PLA/PCL membrane. The duration of degradation of polylactic acid in the human body is about 11 months, and this time is consistent with bone regeneration. [38].



**Fig. 8.** The biodegradation of PLA-PCL, PLA-PCL-HA, PLA-PCL-HA-ZnO, and PLA-PCL-HA-ZnO-Gr membranes into PBS aqueous solution up to 20 weeks.

## **3.3.** Cell compatibility **3.3.1.** *MTT assessment*

The cell/scaffold response of PPH0Z0G0, PPH1Z0G0, PPH1Z1G0, and PPH1Z1G1 nanocomposites was evaluated *in vitro* up to three days. MTT results were shown in Fig. 9 and negative-control was chosen as the one sample which was compared with other membranes. After one day of cell culture, the samples showed

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better cell viability in comparison with the control sample, however there was a significant difference between the control sample and PPH1Z1G1. With increasing in the cell seeding period, the cell viability of membranes and especially those which contain ZnO (1 wt.%) and (Gr wt.%) enhanced significantly than the other samples. Pure PLA and PCL are known as biocompatible materials and it is obvious that PCL or PLA can interact with cells. However, the incorporated HA affects positively on cell/scaffold interaction. Moreover, the incorporation of ZnO and Gr nanoparticles induced cell viability, which may be because of better hydrophilicity as well as being functional co-factor for some enzymes, affecting osteogenic biomarkers by activating the bone regeneration (60, 61). Furthermore,  $Zn^{2+}$  is well-known as an important factor in order to enhancement of healthy bone regeneration (35, 62). Thus, it can be proved that cells can be affected positively by realizing these ions from the nanocomposite.



**Fig. 9.** The cell proliferation and viability of PLA-PCL, PLA-PCL-HA, PLA-PCL-HA-ZnO, and PLA-PCL-HA-ZnO-Gr membranes up to three days; \*  $P \le 0.05$ , \*\*  $P \le 0.01$ , \*\*\*  $P \le 0.001$ , \*\*\*\*  $P \le 0.0001$ .

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## 3.3.2. Antibacterial assessment

The antibacterial performance of PPH0Z0G0, PPH1Z0G0, PPH1Z1G0, and PPH1Z1G1 nanocomposites against S. aureus and E. coli were carried out with zone of inhibition technique. Its results are shown in Fig. 10 and Table 5, showing the antibacterial performances of the nanocomposite coatings. PPH0Z0G0 and PPH1Z0G0 revealed less inhibition of bacterial growth, while PPH1Z1G0 and PPH1Z1G1 showed better performance, with the rates of  $37.97 \pm 0.318$  and  $47.62 \pm 0.571$  mm against S. aureus, and  $38.72 \pm 0.341$  and  $49.4 \pm 0.512$  mm against E. coli, respectively. Regarding the current reports, the antibacterial activities of PPH1Z1G1 consists of not just the direct influence of ZnO nanoparticles but it was improved by zinc ions releasing mechanism as well as producing reactive oxygen species (ROS) (63-65). Moreover, Gr is a storage site for  $Zn^{2+}$ released from ZnO nanoparticles and enhanced the rate of permeability of the cells by having contact with negative bacteria, which finally caused cell deformation, and then leakage. Additionally, some studies have shown that ZnO can lead to lipid and protein peroxidation and further DNA damage. Gr also due to its unique structure could improve the electron transfer rate,

which make this nanocomposite with higher antibacterial activities (66-69).

## 3.3.3. osteoblasts attachment

To observe the cell attachment, the produced optimal nanocomposite were sterilized as follows: the nanocomposite soaked thoroughly solution containing 20 into PBS μg/ml Erythromycin, 3 µg/ml amphotericin B, 20 µg/ml gentamicin, µg/ml 50 penicillin. Then, osteoblasts cells were cultured at 37°C for 24 h, as discussed above. To stabilize the cultured cells, glutaraldehyde (Sigma-Aldrich, USA) was added and the cells were dehydrated by methanol solution. After that, the nanocomposite was treated with tetraoxide osmium for 3 h. as the final step, and was dried before SEM assessment. The field emission scanning electron microscope image is shown in Figure 11. As you can see in Figure 11 the osteoblast cell is completely flattened on the surface. The cell has a philopedia and is completely attached to the surface. In a study by Feng et al., The addition of hydroxyapatite to a polylactic acid polycaprolactone polymeric matrix improved osteoblast adhesion on the matrix surface (70). The results of this study are consistent with the results of research conducted by other researchers.



**Fig. 10.** The schematic anti-bacterial test of the PLA-PCL (a' and a''), PLA-PCL-HA (b' and b''), PLA-PCL-HA-ZnO (c' and c'), and PLA-PCL-HA-ZnO-Gr (d' and d'') membranes against *S. aureus* and *E. coli*, respectively.



| Samples  | S. aureus (mm)    | E. coli (mm)      |
|----------|-------------------|-------------------|
| PPH0Z0G0 | $17.42 \pm 0.32$  | $14.79 \pm 0.219$ |
| PPH1Z0G0 | $19.35 \pm 0.218$ | $20.82 \pm 0.209$ |
| PPH1Z1G0 | 37.97± 0.318      | $38.72 \pm 0.341$ |
| PPH1Z1G1 | $47.62 \pm 0.571$ | $49.4 \pm 0.512$  |

Table 5. A detailed of anti-bacterial performance of PLA-PCL, PLA-PCL-HA, PLA-PCL-HA-ZnO, and PLA-PCL-HA-ZnO-Gr membranes



Fig. 11. The FESEM image relating to osteoblaste cell adhision on the surface of optimum sample.

## 4. CONCLUSION

The current experimental research was accomplished through applying two nanocomposite coating preparation techniques, but aligned in order to optimizing ZnO and Gr nanoparticles loaded into PLA/PCL/HA nanocomposite coating fabricate to а multifunctional membrane for bone defect regenerations. The weight ratio of ZnO and Gr nanoparticles were optimized; according to morphological, mechanical, and chemical characteristics. Consequently, the optimized nanocomposite coatings were considered for further studies to observe bioactivity, release biodegradation, anti-bacterial profile, performance, and cell/scaffold interaction and compared to pure PLA/PCL, and PLA/PCL/HA nanocomposite coatings. The results showed that incorporating ZnO and Gr nanoparticles improved physico-chemical properties as well as

biological performance containing cell viability, proliferation, bioactivity and anti-bacterial activity of produced nanocomposite coatings. The overall results proved that combination of 1 % ZnO and 1% Gr into PLA/PCL/HA membrane could offer a multifunctional platform, a promising alternative for bone disorders and further studies. Given that the Young's modulus of spongy bone is between 0.05 and 0.5 GPa, the optimal matrix is a good option for use in spongy tissue.

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